Kinetic analysis of the temperature dependence of PbSe colloidal quantum dot photoluminescence: Effects of synthesis process and oxygen exposure

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A kinetic model is derived and used to analyze recently published works and new data on the temperature dependence of the spectrally integrated photoluminescence (PL) from PbSe colloidal quantum dot (QDs) with diameters on order 4–6 nm has been studied extensively in solution [1–5] but less so out of solution, where significant applications. 295 K were consistent with only a single radiative transition. PL studies [8] using 5 to 295 K, and detailed lineshape fits of spectra obtained from several batches over the entire temperature range of 5 to 295 K were consistent with only a single radiative transition.

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I. INTRODUCTION

The near-infrared photoluminescence (PL) from PbSe and PbSe/CdSe colloidal quantum dots (QDs) with diameters on order 4–6 nm has been studied extensively in solution [1–5] but less so out of solution, where significant applications also exist [6–11]. One obstacle that has hindered progress in gaining a quantitative microscopic understanding of the optical emission from this material system has been the lack of consistency of the experimental results reported by different groups. Chappell et al. [10] and Hughes et al. [11] recently reported temperature-dependent emission spectra from ≈5-nm-diameter oleate-capped PbSe QDs with apparent contributions from two distinct radiative transitions, the integrated strength of which varies nonmonotonically from 13 to 325 K. Their synthesis is performed in an air-free environment and, depending on procedural details, the temperature dependence of their samples’ emission can be irreversibly sensitive, in drop-cast form, to even minute exposures to air.

In contrast, our previously reported temperature-dependent PL studies [8] using ≈4-nm-diameter PbSe QDs produced in a similar manner [9,12,13], but with less control over the exposure of the samples to air during synthesis, exhibit monotonic, pure Arrhenius-like decay of integrated PL from 5 to 295 K, and detailed lineshape fits of spectra obtained from several batches over the entire temperature range of 5 to 295 K were consistent with only a single radiative transition.

Previously we independently reported that the PL from these samples degraded at room temperature under exposure to oxygen [13], but that this effect was entirely reversible.

These contrasting results motivated us to carry out further experiments on the full temperature dependence of our samples after exposure to air and to develop a kinetic model that might provide insight into to the similarity and differences between these apparently disparate but nominally very similar PbSe QD formulations.

II. SUMMARY OF PHOTOLUMINESCENCE RESULTS

Table I describes a number of PbSe QD samples and relevant treatments reported in four separate publications, along with a different sample used for new oxygen-exposure studies reported below. Figure 1 contains plots of the temperature dependence of the integrated PL strength corresponding to all of the samples identified in Table I.

The integrated QD PL reported by Chappell et al. and Hughes et al., reproduced in Fig. 1, A1–B5, is marked by either strong nonmonotonicity and/or explicit increases with increasing temperature. This is in spite of nominally similar QD syntheses for samples A1 and B1–B5 and no air exposure for samples A1, A2, and B1. Nonmonotonic PL yield persists even after their samples are exposed to a controlled oxygen environment from minutes to hours. This behavior is attributed to PL contributions from two distinct radiative states at different energies in these QD samples, consistent with spectral lineshape analysis, at least in dilute glass-encased samples.
that reveals two peaks separated by several tens of meV, over a range of temperatures.

Independent models \[8,10,11\] have been developed to quantify various aspects of the published PL behavior for subsets of the samples listed in Table I. However, we are unaware of any account of a systematic attempt to fit the entirety of this disparate PbSe nanocrystal temperature-dependent integrated PL data from 5 to 325 K using a single, physically based model.

To inform quantitative comparisons of samples, we describe below a physically based kinetic model that captures the essential behavior of integrated PL data from all samples listed in Table I. The number of free parameters is minimized, and our analysis of the data based on this model highlights the importance of taking into account parameter covariance when extracting reliable fit parameters consistent with the data.

III. MODEL

Various band-structure calculations suggest that the 64-fold degenerate excitonic ground state of bulk PbSe is spread into a manifold of 5\(\times\)S excitonic states in small (3–5 nm diameter) QDs \[14\]. Intervally coupling splits the states over an energy range of tens of meV, while electron-hole exchange splitting further lifts the degeneracies by a smaller amount, on order of a few meV \[14\]. Of these 64 states, only a few have significant oscillator strength.

Assuming that the excitons thermalize to a Fermi-Dirac distribution within the ground-state exciton manifold at all temperatures, the steady-state, spectrally integrated PL emission rate can be expressed as

\[
g_r = \int_{\epsilon_{\text{min}}}^{\epsilon_{\text{max}}} \frac{\gamma_r(\epsilon) f(\epsilon)}{e^{(\epsilon-\mu)/kT} + 1} \, d\epsilon, \tag{1}
\]

where \(f(\epsilon)\) is a sum of 64 \(\delta\) functions defining the energies of the exciton states within the ground state exciton manifold, spanned by \(\epsilon_{\text{min}}\) and \(\epsilon_{\text{max}}\). \(\gamma_r(\epsilon)\) is the radiative decay rate distribution of the states within the manifold, and the remaining part of the integrand is the Fermi-Dirac occupation factor. Assuming weak (linear), nonresonant optical excitation into the high-lying quasicontinuum of the QD absorption spectrum and rapid decay to the ground-state manifold at a rate \(g_{\text{ex}}\), the steady-state quasichemical potential \(\mu\) of the excitons is
FIG. 2. (Color online) Modeled energy-level arrangement. Levels $A$ and $B$ are in thermal quasi-equilibrium and populated by pumping of higher-lying levels at a rate $g_{ex}$. They consist of $N_A$ and $N_B$ states, and the averaged net radiative decay rates from the group of $A$ and $B$ states are $\overline{\gamma}_A$ and $\overline{\gamma}_B$, respectively. The nonradiative decay rate $g_{nr}$, the nature of which is described in detail in the text, is not specific to levels $A$ or $B$ for the purposes of this kinetic modeling, and is indicated by the dashed arrow.

determined by solving

$$g_{ex} = \int_{\epsilon_{min}}^{\epsilon_{max}} \frac{[\gamma_A(\epsilon) + \Gamma_{nr}(\epsilon, T)] f(\epsilon)}{e^{(\epsilon-\mu)/kT} + 1} \, d\epsilon \tag{2}$$

for $\mu$ at each $T$, assuming the other parameters are fixed. Here $\Gamma_{nr}(\epsilon, T)$ is the distribution of nonradiative recombinative rates for different exciton states in the ground-state manifold, which can be temperature dependent.

In order to limit the free parameters in the kinetic model, we make two assumptions: (i) there are at most two states, or more generally two clusters of closely spaced states, that have significant oscillator strengths, and (ii) the net nonradiative decay rate from the entire manifold can be fit using a temperature-dependent function that does not include any level-specific parameters.

These assumptions are schematically summarized in the energy level diagram of Fig. 2. Nonradiative decay in our model is generally from the collective manifold of excitonic states instead of pathways specific to states within the manifold. Assumption (i) is chosen to be consistent with up-to-date observations of thick film PbSe PL, but can be relaxed if future works reveal otherwise. Assumption (ii) does not preclude a more state-specific interpretation of whatever temperature-dependent function is fit to the nonradiative decay rate. The radiative emission rate is thus approximated:

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render their corresponding energies meaningless are omitted, along 80 meV for reasons described in the text. Rates sufficiently small to pathways used to fit to any particular curve was chosen such the number of nonradiative  

\[ \frac{N_B \gamma_B}{N_A \gamma_A} \]

as in Fig. 1 and Table I. Parameter \( \Delta \epsilon_{BA} \) is restricted from above to 80 meV for reasons described in the text. Rates sufficiently small to render their corresponding energies meaningless are omitted, along with those energies.

![Graphs representing parameter ranges](image)

**FIG. 3.** (Color online) Bars represent the ranges of parameter values that, upon substitution into Eq. (7), using Eq. (5), generate or nearly generate the model curves in Fig. 1. Rates are normalized as described in the text, and energies are in meV. Sample labeling is as in Fig. 1 and Table I. Parameter \( \Delta \epsilon_{BA} \) is restricted from above to 80 meV for reasons described in the text. Rates sufficiently small to render their corresponding energies meaningless are omitted, along with those energies.

the procedures described below. The number of nonradiative pathways used to fit to any particular curve was chosen such that \( \min(\chi^2) \), normalized to the number of data points for that curve, was less than three. This threshold was found to admit best fits that captured qualitative behavior of all the sample curves, especially relating to the effects of air exposure.

The sole constraint imposed upon the parameters is an upper limit of 80 meV for \( \Delta \epsilon_{BA} \), based on the spectral range over which the PL was integrated and allowing for possible differences in the Stokes shifts associated with the two radiative levels in sample series A and B. Otherwise, the parameter search domains correspond to the plotting ranges in Fig. 3.

The first segment of our best fit algorithm involves (1) sampling a set of parameters logarithmically from the parameter search domain, (2) applying the Levenberg–Marquardt algorithm with this set of parameters as an initial condition to determine the local best-fit parameters, (3) repeating (1) and (2) a total of \( 10^3 \) times, and (4) associating the best fit of these local best fits with the global best fit. This global best fit was corroborated via application of local minimization to a parameter search domain grid (the latter contrasted with random sampling of the parameter search domain). The logarithm of the likelihood function was found to be quadratic in most parameters near the best-fit points.

Red lines in Fig. 1 are the model curves corresponding to the best-fit parameters found by this procedure. Dotted and dashed lines are plots of the individual model A-state and B-state contributions from each state for the same best-fit parameters.

The second segment of our fit routine—generation of the vertical parameter bars in Fig. 3—consisted of (5) dividing up each parameter search domain into a set of many narrow subdomains, (6) restricting, parameter by parameter of a particular sample, that parameter value to one of those narrow subdomains, (7) sampling a complimentary set of parameters logarithmically from the large parameter search domain, (8) applying the Levenberg–Marquardt algorithm to determine a local best fit, (9) repeating (7) and (8) at total of \( 10^2 \) times, (10) identifying the best fit of those local best fits, (11) checking to see if \( \chi^2 - \min(\chi^2) \leq 4 \) is satisfied for the best of those local best fits, and (12) repeating steps (6) through (11) for every other narrow subdomain. The vertical bars in Fig. 3 are lines that extend from the minimum to maximum parameter values with which some \( \chi^2 - \min(\chi^2) \leq 4 \) exists, as found in steps (5) through (11).

Figure 1 indicates that good quality fits to the data are obtained using the Maxwell–Boltzmann limit of the model. This limit should apply only when \( g_{ex}/N_A \gamma_A \ll 1 \), in which case the results are independent of \( g_{ex} \). We estimated \( g_{ex} \) for both our data sets and those of Chappell et al. using measured laser powers and spot sizes, along with absorption coefficients from the literature [15]. The spontaneous emission rate of the lower emissive state was estimated from lifetime measurements of our drop-cast films and dielectric modeling we previously employed [16]. For both our samples and those of Chappell et al., we estimate \( g_{ex}/N_A \gamma_A \ll 10^{-1} \). Furthermore, if \( g_{ex} \) is increased to the point where the Maxwell–Boltzmann approximation breaks down, it becomes very difficult to fit any data set (impossible in the strongly degenerate limit).

**V. FIT RESULTS**

With reference to the parameter summary in Fig. 3, the most significant difference between the various samples is that high-quality fits required \( N_B \gamma_B/N_A \gamma_A \gg 1 \) for all Chappell et al. and Hughes et al. samples, and \( N_B \gamma_B/N_A \gamma_A \ll 1 \) for all of our samples. This is consistent, in the former case, with the prominent mid-temperature increase in PL and double-peaked mid-temperature spectra, and in the latter case, with our observation of only a single emission peak within the few-meV resolution of previously reported spectra [8]. Also, although a trend in \( N_B \gamma_B/N_A \gamma_A \) with oxygen exposure is admissible, any such trend does not follow necessarily from our kinetic parameters.
analysis, and it is further clear that increasing oxygen exposure does not trend $N_B\Delta E_{\text{BA}} / N_A\Delta E_{\text{BA}}$ in their samples towards the value for our samples.

Fits require the A-B energy separation for series A and B samples to be $\Delta E_{\text{BA}} \approx 55$ to 80 meV. It should be noted that the $\Delta E_{\text{BA}}$ value obtained from fits to the no-exposure data agrees reasonably well with the separation of the emission lines in the solution spectra, but not in the drop-cast samples. This is to be expected since thermalization of the exciton populations is known [8] to have a strong influence on the temperature-dependent Stokes shift of the peak PL emission energy, while there should be minimal effects of thermalization on the spectra obtained in a dilute solution, where the Förster interaction strength is negligible.

For all samples that were not intentionally exposed to oxygen, the integrated PL curves can be fit assuming only a single nonradiative pathway. The associated parameters of this nonradiative pathway (an activation energy around a few tens of meV and normalized prefactor on the order of several to a few tens) are quite similar for the unexposed sample from Chappell et al. (B1) and our unexposed samples (C series and D series excluding D2). Nonradiative decay in series A samples is consistent with a single nonradiative pathway with a considerably lower activation energy. Without information regarding the absolute quantum yield in the case of series A samples, it is not possible from the fitting procedure to separate small differences in assumed activation energy and the corresponding prefactor for this single nonradiative decay process (hence the unbounded arrows for series A samples). Introduction of a second sub-10-meV nonradiative pathway increases fit quality for the Chappell et al. unexposed sample (B1), but the difference in fit quality is small. The slight increase in integrated PL at low temperatures in the A2 sample is not captured with our model, but a third, relatively dark level, just slightly below the A level in energy, could be introduced to capture this behavior.

Samples intentionally exposed to air (B2–B5 and D2) absolutely require the inclusion of two nonradiative pathways for adequate fits. One of these new pathways has a low (sub-10-meV) activation energy, with a corresponding prefactor in the range of 1 to 10, indicated by the red bars in Fig. 3. In our air-exposed sample (D2), the parameters associated with the second nonradiative pathway are not dramatically different than before air exposure, whereas in the case of samples B2–B5, the second pathway has a very high activation energy (>100 meV), and high normalized prefactor (100s of thousands). There is no indication of the need for a high-energy pathway in any of the unexposed or our air-exposed samples. We note that while the low-energy nonradiative pathway is common amongst all air-exposed samples, Chappell et al. report irreversibility of this air-exposure effect for their samples, in stark contrast to the reversibility of the impact of our air exposure (see sample D2 compared to D1 and D2). Finally, we note that inclusion of a third nonradiative pathway with an activation energy of several tens of meV increases the quality of fit for the B2–B5 samples; but again, the improvement is minor.

Regarding phonon-mediation of nonradiative decay, Chappell et al. and Hughes et al. assumed a $N_{\text{ph}}$-phonon absorption functional form for their low-energy, nonradiative pathway [10,11], and they reported fits suggested $N_{\text{ph}} = 4 \pm 2$. In cases where present, if we replace the several-to-ten meV nonradiative Arrhenius pathway ($\Gamma(e^{-\Delta E/kT})$ with a one-phonon emission functional form, $\Gamma'_{\text{ph}}(1 - \exp(-E_{\text{ph}}/kT))^{-1}$, similar-quality fits to those shown in Fig. 1 can be obtained for a phonon energy of $\approx 5$ to 10 meV. From this analysis, there is no preference for the single phonon versus low-activation energy Arrhenius process, but it is nevertheless unambiguous that a nonradiative decay channel with a low (<10 meV) activation energy is required to fit all of their and our oxygen-exposed samples. The lack of preference between these two functional forms can be attributed to substantial parameter covariance.

VI. DISCUSSION

The behavior of the lower-energy A state for samples from series B is very similar to that of the only radiative state that manifests itself in our samples (series C and D). With no oxygen exposure, the integrated PL from the A transition in sample B1 is almost identical to that observed in all series C and D samples save for the one exposed to air (D2). In the case of all oxygen-exposed samples from series B and D, the change in behavior of the A-state contribution is very similar to the change in behavior of our single emissive state in sample D2, up to temperatures above approximately 125 K, beyond which point the A-state contribution in oxygen-exposed B series samples decays more rapidly than in sample D2. It would appear that oxygen introduces a new, low-activation-energy nonradiative decay path for state A, and our single emissive state, which causes more rapid (than in the absence of oxygen exposure) falloff of the emission as temperature, is increased from approximately 5 K. The difference in behavior at higher temperatures for the A-state contribution from samples in series B is apparently due to an additional, high-activation-energy nonradiative pathway that oxygen exposure causes in those samples. It seems plausible that this high-activation-energy nonradiative channel is somehow tied to the B state that is apparent in all series A and B samples, but absent in our series C and D samples.

One explanation for the absence of radiative and nonradiative evidence of the B state in our samples is that the next-highest-lying state with significant oscillator strength in our samples has a $\Delta E_{\text{BA}}$ much larger than $\approx 80$ meV. An alternate explanation for the lack of radiative evidence of the B state in our samples, is the possible impact of nanoparticle shape or symmetry on the oscillator strengths [17] of the states within the ground-state exciton manifold. Positing explanations for these conjectures raises the issue of the nature of both the A and B states; are either or both “intrinsically” excitonic states constructed from the one electron and one hole single-particle states typically calculated for these QDs, or are either or both associated with defects, either bulk like or surface like? Based on most published band-structure calculations for these systems, it is hard to identify an “intrinsically” state that could be associated with the A state in this analysis, since almost all of these calculations conclude that the ground states of the intrinsic manifold are “dark,” with the first “bright” states located on the order of 10 to 20 meV above the intrinsic ground state. Since, with the exception of the Hughes et al.
sample A2, and possibly other Hughes et al. samples similar to the A series, there is no experimental evidence of increasing PL emission from temperatures of 5 K, the A state must be close to the energetically lowest available excitonic state in the QDs, whatever its nature. In the case of the Hughes et al. sample A2, the weak rise in integrated PL at low temperature does suggest that the A state in those samples lies slightly above the ground state, but not by the amount suggested by most published band-structure calculations. Thus, to the extent that current band-structure calculations accurately capture the intrinsic properties of these QDs, this suggests the A state in series B samples, and the only emissive state in our QDs, is a defect state with non-negligible oscillator strength, or that some detail of the nanoparticle structure influences the theoretical oscillator strengths in such a way that low-energy exchange-split states have large oscillator strengths. If the impact of symmetry is sufficient to impact the low-energy A state, it may not be so surprising that the higher-energy B state’s oscillator strength could be quite sensitive to sample synthesis details.

Further experiments and theoretical calculations that include defects and particles of various symmetries are required to explain the nature of the optically active states in PbSe QDs, and the reversibility of the influence of oxygen in some samples, and its irreversible nature in other samples.

VII. CONCLUSION

A kinetic model that allows for luminescent emission from up to two bright clusters of states, and nonradiative decay from any of the excitonic states in the lowest manifold of PbSe QD thin films can accurately reproduce a diverse range of measured temperature-dependent integrated PL emission data from 5 to 325 K. A thorough statistical analysis of the best-fit parameters (at most six, and in some cases as few as two), including their covariance, reveals the commonalities and differences in the optical emission properties of nominally similar PbSe QD samples grown in different laboratories, including their response to oxygen exposure.

One set of samples (labeled C and D) effectively emit from only a single cluster of bright states that have energies at or very close to the lowest of the accessible excitonic states. In the absence of intentional oxygen exposure, the nonradiative decay from these samples is well described by a single Arrhenius process with an activation energy on order of 20 meV. The other sets of samples (labeled A and B) exhibit significant emission from primarily two clusters of states separated by ≈55 to 80 meV, with the lower-energy state very close to the bottom of the manifold of accessible states. In the absence of intentional oxygen exposure, samples A and B either exhibit the same 20 meV activated nonradiative decay behavior of samples C and D, or a nonradiative decay with negligible activation energy, depending on the organic capping layer.

Oxygen exposure of samples B, C, and D changes their behavior in a way that can be described by introducing a second, distinct Arrhenius-like nonradiative decay channel with a relatively low activation energy. The biggest difference between samples B and samples C and D in this regard, is that the effects of oxygen exposure on samples B is irreversible, while it is reversible for samples C and D.

This analysis and sample comparison suggests that the lower-energy transition in samples A and B shares many attributes of the single emissive state in samples C and D. The fact that the energy of this bright state is very close to the bottom of the ground manifold of accessible exciton states is difficult to reconcile with existing calculations of the oscillator strength distribution within the “intrinsic” ground-state manifold of PbSe QDs. This, together with the apparently fickle nature of the higher-energy emissive state evident in one set of samples but absent in the other suggests that either (i) the low-energy emissive state is associated with defects, and/or (ii) that current calculations of band structure and oscillator strengths fail to capture important details, perhaps associated with particle morphology, that have yet to be experimentally determined.

Further experimental specification of the precise physical makeup of these small QDs, and further studies of the impact of oxygen exposure are required to better understand these PbSe QDs.

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